# Application of the Random Pore Model to the Carbonation Cyclic Reaction

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Calcium oxide has been proved to be a suitable sorbent for high temperature  $CO_2$  capture processes based on the cyclic carbonation-calcination reaction. It is important to have reaction rate models that are able to describe the behavior of CaO particles with respect to the carbonation reaction. Fresh calcined lime is known to be a reactive solid toward carbonation, but the average sorbent particle in a CaO-based  $CO_2$  capture system experiences many carbonation-calcination cycles and the reactivity changes with the number of cycles. This study applies the random pore model (RPM) to estimate the intrinsic rate parameters for the carbonation reaction and develops a simple model to calculate particle conversion with time as a function of the number of cycles, partial pressure of  $CO_2$ , and temperature. This version of the RPM model integrates knowledge obtained in earlier works on intrinsic carbonation rates, critical product layer thickness, and pore structure evolution in highly cycled particles. © 2009 American Institute of Chemical Engineers AIChE J, 55: 1246–1255, 2009

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## Introduction

The separation of a pure CO<sub>2</sub> stream, combined with a well-managed geological storage site is considered to be a major mitigation option for climate change. The carbonation reaction serves as the basis for several high temperature CO<sub>2</sub> capture systems when it is coupled with a calcination step to produce a pure CO<sub>2</sub> stream. The use of the carbonation-calcination loop of CaO/CaCO<sub>3</sub> is now accepted as a viable technique for the capture of CO<sub>2</sub> in postcombustion<sup>2–5</sup> or in precombustion<sup>6–10</sup> routes. Figure 1 shows a possible scheme for the proposed CO<sub>2</sub> capture process for postcombustion application. The system mainly consists of two interconnected circulating fluidized bed reactors: a carbonator and a regenerator or calciner. In the carbonator, the CO<sub>2</sub> present in

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the flue gas stream coming from a boiler meets a flux of CaO and reacts to form CaCO<sub>3</sub>. In the calciner, a secondary fuel is fired with pure oxygen to supply the necessary heat to calcine the CaCO<sub>3</sub> formed in the carbonator and decompose it into CaO (which is returned to the carbonator) and CO<sub>2</sub>, suitable for final purification, compression, and geological storage.

The design of the carbonator reactor requires a good understanding of the carbonation reaction rates of the CaO particles that enter the carbonator from the calciner. High reaction rates between the CO<sub>2</sub> in the flue gas and the sorbent particles are necessary to design absorbers of a reasonable size. Fresh calcined lime is known to be a very reactive solid in the carbonation reaction. However, the average sorbent particle must undergo many carbonation-calcination cycles<sup>3</sup> as a result of which sorbent capture capacity will decrease rapidly. Since the solids in both reactors in Figure 1 are assumed to be well mixed, the make up flow of limestone and the purge of CaO will result in a wide distribution of

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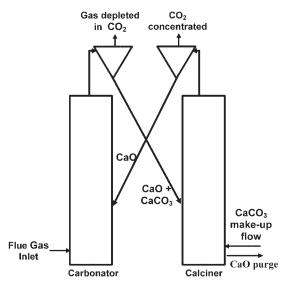


Figure 1. Schematics of the carbonation-calcination

particle types in terms of cycle number and reactivity. Previous studies on the reversibility of the carbonation/calcination reaction have shown that carbonation is far from reversible in practice.<sup>2,4,6,11–16</sup> After a fast, chemically controlled initial reaction stage, a second slower reaction stage controlled by the diffusion in the product layer, CaCO<sub>3</sub>,takes place. 13 It has also been observed that the transition between the fast and slow regimes takes place quite suddenly at a given level of conversion, and that this level of conversion decreases as the number of carbonation/calcination cycles increases. The evolution of the sorbent capture capacity of natural sorbents, with the number of cycles and varying process variables, has been studied in previous works. 14,15 It was found that this capacity decreases rapidly in the first 20 cycles but tends to stabilize at very high cycle numbers<sup>15</sup> at a residual conversion. Calcination temperatures over 950°C, and/or extended calcination times accelerate the degradation of the sorbent, which therefore attains the residual capture capacity with a lower number of cycles. Detailed observation by mercury porosimetry and SEM of the textural changes in limestone during cycling 14,16,17 led to the conclusion that the main mechanism of sorbent deactivation is the progressive sintering, or grain growth, of the originally rich texture resulting from the first calcination. According to this mechanism, after a certain number of cycles, the CaCO<sub>3</sub> formed during carbonation will occupy all the space in the small pores plus a small fraction of the large voids. However, the occupation of the large pores is limited by the thickness of the product layer that marks the onset of the slow carbonation period. Secondorder effects (pore mouth blockage, isolated voids in the calcined particle, particle shrinkage) can also be detected in some special sorbents and conditions. 17 Essentially, though, it is the grain sintering mechanism as the number of cycles increases, and the thickness of the product layer in the large voids, that marks the decay of capture capacity.

Different descriptions of the carbonation reaction have been proposed through a gas-solid reaction model, 2,8,13,18,19 and a review of these carbonation models has been recently carried out by Stanmore and Gilot.<sup>20</sup> Among the models pro-

posed are semiempirical models, 2,18 where the carbonation reaction is described by a simple rate expression based on the difference between the maximum carbonation conversion and the conversion at any point, multiplied by the kinetic constant and the average CO2 concentration in the reaction atmosphere. Johnsen et al.<sup>8</sup> applied the shrinking core model to describe the carbonation reaction based on their SEM/EDS observations. They observed that the oxygen profiles in a partially carbonated particle displayed a nonuniform reaction pattern. The particle had an external rim indicating a high concentration of CaCO<sub>3</sub> and an inner core with a higher concentration in CaO. A third group of researchers applied pore models. <sup>13,19</sup> These pore models describe the carbonation reaction with greater accuracy as they take into account the evolution of pore size distribution during the cycling. Furthermore, many SEM studies of highly cycled samples suggest that a pore model may provide a better geometrical approximation for describing the texture of calcines. Adjusting the pore model to the grain model, with a suitable correction of the structural parameters, was established by Bhatia and Perlmutter as an effective procedure when they compared the grain model with their random pore model (RPM). The RPM model considers the pore structure as a network of randomly interconnected pores, defines the particle structural parameters based on this geometry and finally, when it is applied to the carbonation reaction, provides kinetic parameters by fitting the experimental reaction rate data obtained from "fresh lime"13 (only one calcination). Recently, a new gas-solid model based on discrete pore size distribution measurements was developed by Sun et al. 19 This model uses the initial pore size distribution of the calcine as input data, the only fitting parameter being the effective diffusivity in the product layer (which is also dependent on the evolution of the pore system). The same group determined intrinsic kinetic constants in a previous work and they included them in the model equations.<sup>23</sup> The results obtained from the application of their model are in agreement with experimental data over a wide range of temperatures and partial pressures of CO<sub>2</sub>. However, this model is difficult to apply because it requires as input data the initial pore size distribution after calcination. Until now this model has only been applied to "fresh lime."

The need for reaction rate models to the carbonation of highly cycled CaO particles is relatively new procedure and related to the development of carbonate looping processes for CO<sub>2</sub> capture. In a previous work by our group, <sup>24</sup> a simple homogenous model was applied to describe the carbonation reaction during the fast reaction period of highly cycled particles of CaO. In these conditions, all resistance to chemical reaction was eliminated due to the distribution of wide pores resulting from multiple carbonation/calcination. 16,17 A simplified homogeneous reaction model was found to be sufficient for describing the carbonation reaction of highly deactivated particles.

We have attempted in this work to derive a carbonation reaction model valid for any number of cycles and able to represent CaO conversion with time as a function of the cycle number and carbonation reaction conditions (temperature, CO<sub>2</sub> partial pressure). To achieve this objective, the following experimental work was carried out in order to obtain information about the reactivity of different types of lime particles toward carbonation.

## **Experimental**

The cyclic carbonation and calcination reactions were experimentally studied in a thermogravimetric analyzer (TGA) specially designed to derive reactivity data during carbonation in long multicycle carbonation-calcination tests. The TGA, consisted of a quartz tube placed inside a twozone furnace capable of working at temperatures up to 1000°C. The temperature and sample weight were continuously recorded on a computer. The reacting gas mixture (CO<sub>2</sub>, O<sub>2</sub>/air) was regulated by mass flow controllers and fed into the bottom of the quartz tube. A special characteristic of this TGA is the presence of two zones in the furnace capable of working at different temperatures. The furnace can be moved up and down by means of a pneumatic piston. Its position with respect to the platinum basket alternates between calcination conditions (>850°C) or carbonation conditions (around 650°C). For each run in the TGA around 2 mg of sorbent was introduced in the sample holder to avoid interparticle diffusion effects in the platinum basket. Preliminary experiments were carried out to determine the total gas flow needed to eliminate external diffusion effects around the sample basket (this was finally set to  $4 \times 10^{-6}$ m<sup>3</sup>/s, i.e., about 0.06 m/s of superficial gas velocity around the sample basket at 650°C and 0.08 m/s at 950°C). Preliminary experiments were performed using an empty sample holder and an inert material to determine the effect of disturbances on the weight readings when the furnace is moved from the calcination to the carbonation position (a rapid change in temperature modifies the gas velocity around the sample basket). After correcting the data obtained from the preliminary blank tests, CaO conversion vs. time for each cycle was calculated by measuring the weight losses and assuming that the CaO was converted to CaCO3 during carbonation. At the end of each run, the samples were weighed in a different balance to check the accuracy of the TGA experiment. Good agreement was found in all cases between the overall conversion calculated from this final weight difference and the TGA conversion. Although the TGA has been designed to allow for fast changes in temperature around the sample holder, there is still a delay in the order of 30-60 s before the desired carbonation temperature is reached after the calcination step. To stop the carbonation reaction during

this temperature stabilization period, the flow of CO2 was switched off until the carbonation temperature was stable within a ±5 K difference with respect to the preset temperature. Two different limestones were used during the experiments (a Spanish limestone called Imeco and a Polish limestone called Katowice). A series of test was carried out to determine the effect of particle size on the carbonation rate experiments. Three particle size cuts were tested:  $50-75 \mu m$ , 100-300  $\mu$ m, and 300-600  $\mu$ m. However, since the main objective of this work was to derive intrinsic carbonation rates, a narrow particle size cut was used (50–75  $\mu$ m) for both limestones in most tests. Both limestones were characterized by using a Hg Porosimeter Quantachrome PoreMaster to determine their structural parameters according to the RPM. The carbonation reaction was studied over the range of 550 to 750°C and the pCO<sub>2</sub> was varied between 0.01 and 0.1 MPa.

### Results

Figure 2 shows typical curves representing CaO conversion, *X*, vs. time for different carbonation temperatures and for both limestones (weight changes were measured every second). The curves represent key features described in other published works related with the carbonation reaction. <sup>2,6,11–20,23,24</sup> After a fast initial reaction controlled by chemical reaction (kinetic regime), there is an abrupt change in the carbonation reaction rate. This change to a second slower stage has been attributed to the mechanism of CO<sub>2</sub> diffusion through a CaCO<sub>3</sub> layer that starts to control the reaction at this point. <sup>13,16,19</sup>

Alvarez and Abanades<sup>16</sup> concluded that the thickness of the product layer was a critical parameter for understanding the carbonation reaction during the fast and slow reaction periods and the transition between these two differentiated reaction rates. By means of mercury porosimetry they observed that, for several limestone types and for a different number of carbonation/calcination cycles, a carbonate layer of 49 nm formed on the available CaO surface and that this was responsible for the end of the fast reaction period. However, characterization of this critical parameter does not provide sufficient information to ensure the correct design of the carbonator reactor. It is also necessary to establish the correct carbonation rates before the formation of this critical

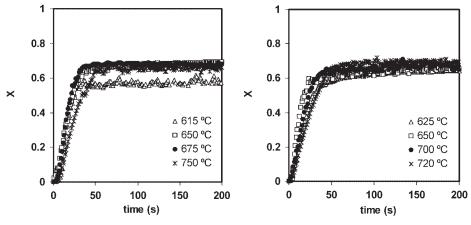


Figure 2. CaO conversion curves for different carbonation temperatures.

(Left) Imeco limestone. (Right) Katowice limestone. Calcination in air at 900°C, 10 min: pCO<sub>2</sub> during Carbonation 0.01 Mpa.

Table 1. Characteristics of the Calcines from the Limestones Tested in This Work and Structural Parameters

Limestone	$S_{\rm o}~({\rm m}^2/{\rm m}^3)$	$L_{\rm o}~({\rm m/m^3})$	3	Ψ	$k_{\rm so}~({\rm m}^4/{\rm kmols})$	$E_{\rm aK}$ (kJ/kmol)	$D_{\rm o}~({\rm m}^2/{\rm s})$	$E_{\mathrm{aD}}$ (kJ/kmol)
Imeco Katowice	$42 \times 10^6$ $35 \times 10^6$	$4.16 \times 10^{14} \\ 2.63 \times 10^{14}$	0.47 0.46	1.52 1.41	$0.559 \times 10^{-5} \\ 0.529 \times 10^{-5}$	$21.3 \times 10^{3} \\ 19.2 \times 10^{3}$	$3.37 \times 10^{-6}$ $4.32 \times 10^{-6}$	$163 \times 10^3$ $163 \times 10^3$

Kinetic and diffusion parameters defined further on in the article are also included in this table.

product layer. The carbonation reaction rate before the formation of the critical product layer will probably have a component of an intrinsic kinetic nature and a component associated to the diffusivity of CO2 through the growing carbonate layer until the small pores are filled up to the critical value of ~50 nm, after which diffusivity tends to be very low. This last point is consistent with the findings of Mess et al. 25 who found that CO<sub>2</sub> diffusivity in the carbonate product layer decreased rapidly with increasing carbonate conversion. Therefore, our first application of the RPM model to the experimental results was aimed at determining the intrinsic parameters characteristic of a reaction regime somewhere in between the two extremes described earlier: kinetic control for low carbonation conversions and increasing importance of diffusion control as the product layer of CaCO<sub>3</sub> builds up. In fact, the RPM has already been developed and applied in the past for this purpose, and for other gas-solid reactions 13,21,22,26 and so only needed to be adapted to the special characteristics of the calcines obtained from the multiple calcination-carbonation cycles.

Bathia and Perlmutter developed a general expression for the instantaneous solid–gas local reaction rate applicable to porous systems in the presence of product layer diffusion resistance:

$$\frac{dX}{dt} = \frac{k_{\rm s}S_{\rm o}C(1-X)\sqrt{1-\Psi\ln(1-X)}}{(1-\varepsilon)\left[1+\frac{\beta Z}{\Psi}(\sqrt{1-\Psi\ln(1-X)}-1)\right]} \tag{1}$$

This expression correlates particle conversion, X, with the internal solid pore structure by defining the structural parameter  $\Psi$ , which accounts for this internal particle pore structure in terms of:

$$\Psi = \frac{4\pi L_0 (1 - \varepsilon)}{S_0^2} \tag{2}$$

where  $L_{\rm o}$  represents the initial total pore length in the porous system per unit of volume;  $S_{\rm o}$  the initial surface area per unit of volume and  $\varepsilon$  the porosity.  $L_{\rm o}$ ,  $S_{\rm o}$ , and  $\varepsilon$  can be directly calculated from the mercury porosimetry data<sup>26</sup> as follows:

$$S_{\rm o} = 2 \int_{0}^{\infty} \frac{v_{\rm o}(r)}{r} dr \tag{3}$$

$$L_{\rm o} = \int_{0}^{\infty} \frac{v_{\rm o}(r)}{\pi r^2} dr \tag{4}$$

and

$$\varepsilon = \int_{0}^{\infty} v_{\rm o}(r) dr \tag{5}$$

Mercury porosimetry was applied to fresh calcines of both limestones in order to determine these parameters. The results are compiled in Table 1 together with the structural parameter  $\Psi$ . Figure 3 shows the pore size distribution for the first calcines of both limestones. They both present a unimodal pore distribution although the Katowice limestone shows a wider pore distribution. The Imeco limestone has a mean pore diameter of 35 nm, whereas the Katowice limestone presents a mean pore diameter of  $\sim$ 50 nm.

The carbonation reaction was found to be first order with respect to the  $CO_2$  concentration up to  $pCO_2$  0.1 MPa<sup>2,13,24,27</sup> (see also Figure 4). This is in contrast with the results of Sun et al.<sup>23</sup> who considered the reaction to be first order up to 0.01 MPa and zero order for higher  $CO_2$  partial pressures.

Radial diffusion resistances through the pore network of the particle were neglected in our application of the RPM model to the experimental data. Carbonation tests conducted with different particle size (Figure 5) up to what can be reasonable particle size of these materials when used in similar circulating fluidized beds combustors (typically lower than 300  $\mu$ m) shows that its effect is modest on reaction rates. Furthermore, Figure 5 shows that the effects are almost negligible after the fourth carbonation cycle. Since the application of carbonate looping cycles in postcombustion processes will require operation with fine and highly cycled materials in CFBs, we have focused the application of the RPM model on the 50–75  $\mu$ m size cut.

For a reversible first-order system, the general rate expression from Eq. 1 can be simplified and integrated in the regime of chemical reaction control, and therefore the particle conversion is given by:

$$\frac{1}{\Psi} \left[ \sqrt{1 - \Psi \ln(1 - X)} - 1 \right] = \frac{k_{\rm s} S_{\rm o} (C_{\rm b} - C_{\rm e}) t}{2(1 - \varepsilon)} \tag{6}$$

This expression was used to derive intrinsic kinetic constants for the carbonation reaction. To obtain reliable parameters

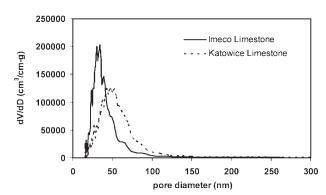


Figure 3. Pore size distribution for both limestones tested.

Calcination at 900°C in air.

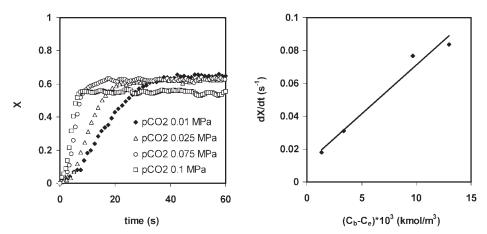


Figure 4. (Left) Experimental carbonation curves for different CO<sub>2</sub> partial pressures; carbonation temperatures 650°C, Imeco limestone; (right) mean delta conversion with time for different CO2 partial pressures.

from this fitting exercise, it was important to ensure that external diffusion resistances were eliminated (by adjusting the thermo balance to allow a sufficient gas flow around the sample basket, and using a small amount of sample as indicated in the Experimental section). Figure 6 shows a sample of the results, from which the left hand side in Eq. 6 has been plotted against time. Therefore from the initial part of the slope in Figure 6,  $k_s$  for each experiment can be calculated.

An Arrhenius representation was plotted for individual values of  $k_s$  obtained from the conversion curves at different temperatures. From the slope and the ordinate in the origin,  $k_{so}$  (m<sup>4</sup>/kmols) and  $E_{aK}$  (kJ/kmol) were determined for both sorbents.

$$k_{\rm s} = k_{\rm so} \exp(-E_{\rm aK}/RT); k_{\rm s} ({\rm m}^4/{\rm kmols})$$
 (7)

The values of  $k_{\rm so}$  and  $E_{\rm aK}$  have been included in Table 1 for both limestones tested. The mean activation energy for the carbonation reaction,  $(20.3\pm 1.0) \times 10^3$  kJ/kmol, is in close agreement with the values found by Sun et al.,  $(29 \pm 4) \times 10^3$ 

kJ/kmol for limestones, and fairly consistent with the values extracted from the equilibrium data,<sup>23</sup> although there are discrepancies with the values presented by Dennis and Hayhurst<sup>27</sup> and Bhatia and Perlmutter, 13 who reported that the carbonation reaction had zero activation energy in the kinetically controlled region. However, the  $k_s$  values obtained in the range of temperatures tested fall within the range of values reported by these authors (mean value of  $0.05 \times 10^{-5}$  m<sup>4</sup>/kmols against the  $0.06 \times 10^{-5}$  m<sup>4</sup>/kmols obtained by Bhatia and Perlmutter for the same range of temperatures).

Within the limit of product layer diffusion control,  $k_s/D_p$ tends to infinity and Eq. 1 can be integrated to the limiting

$$\frac{1}{\Psi} \left[ \sqrt{1 - \Psi \ln(1 - X)} - 1 \right] = \frac{S_o}{(1 - \varepsilon)} \sqrt{\frac{D_p t}{2Z}} \tag{8}$$

 $D_{\rm p}$  being the apparent product layer diffusion coefficient. This can be estimated representing the left hand side in Eq. 8 vs. the root of time, as shown in Figure 7. This parameter is

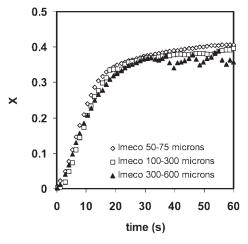


Figure 5. Experimental carbonation curves for different particle sizes.

Fourth carbonation cycle; Carbonation temperature 650 °C, pCO<sub>2</sub> 0.01 MPa.

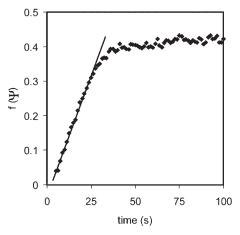


Figure 6. Representation of  $f(\Psi)$  vs. time for an individual curve from Figure 2.

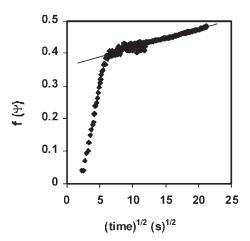


Figure 7. Representation of  $f(\Psi)$  vs. the root of time for an individual curve from Figure 2.

also related to the effective diffusion coefficient, D, as follows:

$$D_{\rm p} = \frac{bM_{\rm CaO}DC}{a\rho} \tag{9}$$

From the slopes in each figure (see Figure 7), the apparent diffusion coefficient,  $D_p$ , and therefore the effective product layer diffusion coefficient, D, have been estimated. This parameter can also be represented in an Arrhenius equation 13:

$$D = D_0 \exp(-E_{aD}/RT); \text{ with } D(\text{m}^2/\text{s})$$
 (10)

The values for  $D_{\rm o}$  and  $E_{\rm aD}$  are compiled in Table 1 and the mean activation energy,  $E_{\rm aD}$  (163  $\pm$  15)  $\times$  10<sup>3</sup> kJ/kmol, is in close agreement with the values reported by Bhatia and Pelmutter<sup>13</sup> and compiled by Mess et al.  $^{25}$  (178 × 10<sup>3</sup> kJ/kmol for temperatures above 515°C).

At this point, the intrinsic parameters  $k_s$  and  $D_p$ , for the carbonation reaction were determined and the carbonation conversion with time can be calculated using the RPM models (Eq. 1). However, we found that if the product layer diffusion resistance is calculated from the very beginning of the reaction much lower CaO conversions than experimental ones are obtained (despite the low thickness of the product layer at low carbonation conversions). Furthermore, the abrupt change in conversion rate that is apparent in the experimental plots of conversion vs. time (see for example Figure 2) is not sufficiently well reflected by the RPM model when diffusion coefficients lower than those compiled in Table 1 are taken into consideration in order to model the diffusion resistance at low carbonation conversions. It is therefore more practical to distinguish two reaction periods in the model: a first period where diffusion through the growing product layer is totally neglected (kinetic control only, the progress of reaction is governed by Eq. 14) and a second period where diffusion resistance is added to kinetic resistance (Eq. 16). The change in reaction rate between these two periods can be justified via two mechanisms: the effect of a variable product layer diffusion resistance that increases, as the product layer thickness increases (this was confirmed in the case of Mess et al.<sup>25</sup> for thick product layers) or alternatively, the initial formation of CaCO3 islands on the free surface of CaO which grow until the individual CaCO3 grains coalesce to form a CaCO3 product layer that is able to seal the free CaO surface. In both mechanisms, there is reason to believe that diffusion reaction control starts to take effect (through the effective CO<sub>2</sub> diffusivity of Eq. 10) only after a certain level of product layer thickness has been reached. In both mechanisms, CO2 diffusion is very fast and does not affect the progress of carbonation for low conversions (low product layer thickness). From a practical point of view, both mechanisms can be simplified and represented in the same form: an initial reaction stage controlled by the kinetic constant and a second stage where both the diffusion coefficient and kinetic constant control the progress of the carbonation reaction.

In the following paragraphs we will try to define the threshold that divides the two reaction regimes. Some published works (Sun et al.<sup>23</sup>) have limited the kinetic control to the initial stages of reaction (only up to a very low carbonation conversion). However, our experimental data show that the fast stage of the carbonation reaction follows an almost linear slope up to a CaO molar conversion of around 0.5-0.6 for a "fresh" calcined sample, when the maximum carbonation conversion is around 0.7. Therefore, the threshold between kinetic control and kinetic and diffusion control must be higher.

To determine this threshold between kinetic control and combined product layer diffusion and kinetic control, we resort again to the pore geometry distribution implicit in the RPM model, which was used earlier by Alvarez and Abanades<sup>16</sup> to calculate the product layer thickness at the end of the fast carbonation reaction (which included both the kinetic period and the period of transition to a very low reaction rate controlled only by diffusion). According to this model, for a given pore size distribution, the evolution of particle conversion with the product layer thickness can be

$$X = \frac{M_{\text{CaO}}}{V_{\text{CaCO}_3}^{\text{M}}} \sum_{i} \left[ \left( \frac{R_i + h_i - \delta_i}{R_i} \right)^2 - \left( \frac{R_i - \delta_i}{R_i} \right)^2 \right] \Delta V_i^{\text{cum}} \quad (11)$$

where  $\Delta V^{\text{cum}}$  is the delta volume from direct Hg mercury porosimetry,  $R_i$  is the pore radius for every pore interval,  $\delta_i$ corresponds to the decrease in pore radius upon carbonation, and  $h_i$  is the product layer thickness at every pore interval. The decrease in pore radius during carbonation was estimated from geometrical considerations and can be related to pore radius, product layer thickness, and  $\alpha$ , according to:

$$\delta_{\rm i} = h\alpha + R_{\rm i} - \sqrt{h^2\alpha^2 + R_{\rm i}^2 - \alpha h^2}$$
 (12)

The parameter  $\alpha$  corresponds to the volume fraction of carbonate invading the former volume of pore i, and can be calculated from the molar volumes of CaO (16.9  $\times$  10<sup>-3</sup> m<sup>3</sup>/ kmol) and CaCO<sub>3</sub> (36.9  $\times$  10<sup>-3</sup> m<sup>3</sup>/kmol) according to:

$$\alpha = \frac{V_{\text{CaCO}_3}^{\text{M}} - V_{\text{CaO}}^{\text{M}}}{V_{\text{CaCO}_3}^{\text{M}}} = 0.54$$
 (13)

When the small pores are filled up, the product layer thickness becomes:  $h_i = \min(\sqrt{\frac{1}{\alpha}}R_i, h_L)$ ;  $h_L$  being the critical product layer thickness found in Alvarez and Abanades<sup>16</sup> (49 nm).

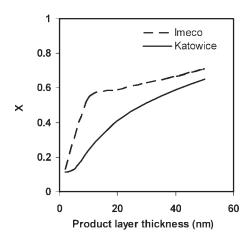


Figure 8. Evolution of particle conversion with the growth of product layer thickness on pore size distribution from Figure 3.

If this model is applied to the pore distributions from Figure 3, the CaO conversion from this point can be obtained for different product layer thicknesses. The results are plotted in Figure 8. It can be seen from Figure 3 that the mean pore diameter for the Imeco limestone is 35 nm, and that it presents a narrow pore size distribution. Therefore, by applying Eqs. 11-13 it can be observed in Figure 8 that CaO conversion rapidly increases while the smaller pores are being filled and that later the product layer continues to grow in the bigger pores until it reaches the limit of 49 nm established in the previous work cited earlier. 16 The Katowice limestone mean pore diameter is 50 nm, and its pore size distribution is wider than for Imeco limestone. Therefore, when the pore model developed by Alvarez and Abanades is applied, the CaO conversion curve and also the transition between the filling of the small pores and the larger ones is more gradual. According to the experimental conversion curves represented in Figure 2 and the relation between the product layer and particle conversion (Eq. 11), a mean product layer thickness of between 30 and 40 nm is determined as the borderline between kinetic and combined kinetic and diffusional control.

In the light of the previous analysis, the application of the RPM to the carbonation reaction was carried out as follows:

Assuming that a chemically controlled reaction stage governs the CaO conversion (up to a product layer of 30–40 nm), Eq. 14 was applied to calculate the CaO conversion as a function of time thus:

$$X = 1 - \exp\left(\frac{1 - \left(\frac{\tau}{2}\Psi + 1\right)^2}{\Psi}\right) \tag{14}$$

where

$$\tau = \frac{k_{\rm s}(C_{\rm b} - C_{\rm e})S_{\rm o}t}{(1 - \varepsilon)} \tag{15}$$

When the product layer thickness reaches 30–40 nm, and the particle conversion is around 0.55–0.6 for the first cycle (this value of conversion corresponding to the transition between regimes will be named  $X_{\text{K-D}}$ ), diffusion resistance increases and both the diffusion coefficient and kinetic constant control

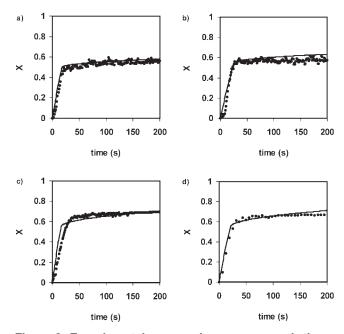


Figure 9. Experimental conversion curves and those predicted by the RPM for Imeco Limestone.

Carbonation temperature: (a) 550  $^{\circ}$ C; (b) 615  $^{\circ}$ C; (c) 650  $^{\circ}$ C and (d) 715  $^{\circ}$ C. pCO $_2$  during carbonation 0.01 MPa. Reaction and structural parameters from Table 1.

the carbonation reaction. The CaO conversion from this point can then be calculated from Eq. 16.

$$X = X_{k-D} + \left(1 - \exp\frac{1}{\Psi} - \frac{\left[\sqrt{1 + \beta Z\tau} - (1 - \frac{\beta Z}{\Psi})\right]^2 \Psi}{\beta^2 Z^2}\right)$$
(16)

$$\beta = \frac{2k_{\rm s}a\rho(1-\varepsilon)}{M_{\rm CaO}bD_{\rm p}S_{\rm o}}$$
 (17)

Figures 9 and 10 show the results obtained from the application of Eqs. 14-17 using the structural, kinetic, and

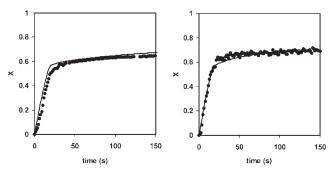


Figure 10. Experimental conversion curves and those predicted by the RPM for Katowice Limestone.

Carbonation temperature: (left) 600 °C; (right) 650 °C. pCO $_2$  during carbonation 0.01 MPa. Reaction and structural parameters from Table 1.

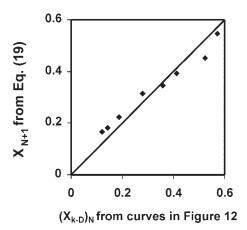


Figure 11. Experimental  $X_{k-D}$  from Figure 12 vs.  $X_{N+1}$  from Eq. 19 for Imeco limestone.

diffusional parameters from Table 1. As can be seen, the results reflect fairly well the experimental results for the range of temperatures tested and for the timescale considered suitable for the  $\rm CO_2$  capture process proposed. Similar curves were plotted for the Katowice limestone.

At this point, the RPM has been successfully applied to describe the carbonation of the CaO particles resulting from one calcination. However, it is well known that the sorbent texture evolves with the number of carbonation/calcination cycles and that the slow carbonation period begins at a lower CaO conversion. Because of repetitive carbonation/calcination cycles, the initial rich texture of the fresh calcines evolves to a larger pore size distribution, and these results in a loss of surface for the reaction. To apply the model described in this work to a multicycle carbonation process, it is necessary to assume that the CaCO3 product layer on which the change of reaction mechanism takes place (from kinetic to kinetic plus diffusion control) is similar to the one obtained in the previous paragraphs for the first cycle. Ideally we should know the pore size distribution after every calcination (to obtain  $S_N$  and  $L_N$  as in Eqs. 3 and 4) and apply the model described through Eqs. 11-13. But this would require a detailed sintering model to be able to estimate the pore size distribution during cycling. Although no such model exists, on the basis of the knowledge acquired on the evolution of the carbonation reaction with the number of cycles and the textural evolution of the sorbent, a series of simplifications can be performed to calculate the structural parameter of the sorbent with the number of cycles and  $(X_{\text{K-D}})_N$ . Since large pore diameters (200–800 nm) are typical of highly cycled samples, it can be assumed that the CaCO<sub>3</sub> forms a layer (up to the limit of thickness  $h_{\text{L}}$ , ~49 nm) on the reaction surface  $S_N$  (m²/m³) which can therefore be estimated as follows for highly cycled particles of large pores (pore diameters [dmt] 50 nm)<sup>24</sup>

$$S_N = S_0 X_N (\text{m}^2/\text{m}^3) \tag{18}$$

where  $X_N$  is the maximum carbonation conversion that the CaO particles achieve during the cycle number, N.

$$X_N = \frac{1}{\frac{1}{(1 - X_r)} + kN} + X_r \tag{19}$$

For a large number of limestones and carbonation and calcination conditions,  $^{15}$   $X_{\rm r}$  was found to be equal to 0.075 and k to be equal to 0.52. Assuming that the total pore volume will remain almost constant with the number of calcinations and that the pore diameter increases in size up to 1000 nm for highly cycled particles,  $L_N$  can be estimated thus:

$$L_N = L_{\rm o} X_N \frac{rp_{\rm o}}{rp_N} ({\rm m/m^3}) \ rp_{\rm o}/rp_N \approx 1 {\rm for \ the \ initial \ cycles};$$
  
 $rp_{\rm o}/rp_N \approx 0.1 {\rm for \ highly \ cycled \ particles} \ (20)$ 

It can be experimentally observed (see Figures 12 and 13) that there is still an abrupt change in reaction rate at a given CaO conversion (named  $X_{K-D}$  in this work) for highly cycled samples, whose value decreases with the number of cycles. This conversion at which the transition between kinetic and combined diffusion takes place, must also be related to the mean product layer ( $h_{K-D}$ ):

$$(X_{K-D})_{N} = \frac{S_{N} M_{CaO} h_{K-D}}{V_{CaCO}^{M} \rho_{CaO}}$$
(21)

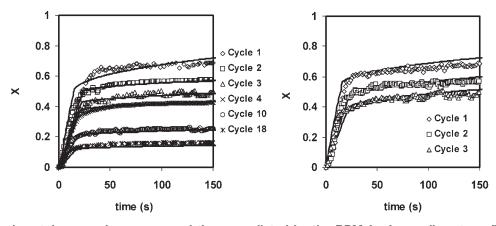


Figure 12. Experimental conversion curves and those predicted by the RPM for Imeco limestone (left), and Katowice limestone (right) for different cycle number.

Carbonation temperature 650°C, pCO<sub>2</sub> during carbonation 0.01 MPa. Reaction parameters from Table 1.

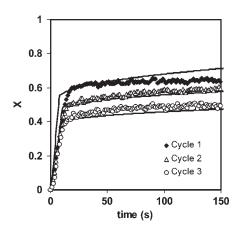


Figure 13. Experimental conversion curves and those predicted by the RPM for Imeco limestone applied for different cycle numbers. pCO2 during carbonation 0.025 MPa.

Carbonation temperature: 640 °C. Reaction parameters from Table 1.

Using the experimental  $X_{K-D}$  values from the individual curves in Figures 9 and 10, values of product layer thickness ranging from 30 to 42 nm were found to determine the change in reaction mechanism  $(h_{K-D})$ . From these results, and for a practical application of the model, a mean value of 38 nm is proposed as the threshold between kinetic and combined kinetic plus diffusional control for any

It has also been experimentally observed that  $X_{K-D}$  for a given cycle N is in close agreement with the maximum capture capacity X for cycle N + 1 calculated through Eq. 19. As can be seen in Figure 11,  $(X_{K-D})_{N,\text{experimental}}$  $\approx (X)_{N+1}$  from Eq. 19. It is not yet clear if this is a relevant observation for understanding the sintering mechanism during the carbonation-calcination cycles, but in the absence of better information this rule could be applied to estimate  $X_{K-D}$  for other limestones and conditions that produce deactivation curves different to those used in this work.

Once  $S_N$ ,  $L_N$  (and therefore  $\Psi_N$ ), and  $X_{K-D}$  have been estimated for every cycle, Eqs. 14-17 can be applied to calculate particle conversion against time. Figure 12 shows the results obtained with the carbonation model for different numbers of cycles and for both limestones tested.

The model described earlier has also been applied to obtain conversion curves vs. time for experiments carried out with the Imeco limestone, for carbonation  $T = 650^{\circ}\text{C}$ and  $pCO_2 = 0.025$  MPa in air. Figure 13 shows the evolution of the carbonation reaction after three successive calcination-carbonation cycles.

As can be seen from the previous figures, the RPM model proposed here is capable of fitting with a reasonably good accuracy the carbonation rates obtained in the relevant range of reaction times and conditions for the carbonate looping concept to be successfully applied.

# **Conclusions**

The RPM has been applied in this study to estimate the intrinsic kinetic and diffusion parameters for the carbonation reaction in CaO particles that have experienced one or more calcination-carbonation cycles. The values for the parameters estimated are in close agreement with values found in the literature for carbonation after one calcination. The carbonation reaction can be successfully described as a two stage reaction. In the first stage, the carbonation is controlled by chemical reaction and in the second period there is a combined control by chemical reaction and CO2 diffusion through the product layer. The RPM can be applied to determine the thickness of the product layer that marks the change in the reaction mechanism. This change was found to be around 30-42 nm for the limestones tested for different conditions and cycle numbers, with a mean value of 38 nm. Below this product layer thickness the particles react following a chemically controlled, first-order reaction, with an activation energy of  $21.3 \times 10^3$  kJ/kmol for the Imeco, and  $19.2 \times 10^3$  kJ/kmol for the Katowice limestone. The pre-exponential factors were  $0.559 \times 10^{-5}$  and  $0.529 \times$ 10<sup>-5</sup> m<sup>4</sup>/kmols, respectively. At higher conversions, the additional resistances associated with the CO<sub>2</sub> diffusion mechanism (mean activation energy,  $E_{\rm aD}$  being  $163 \times 10^3$ kJ/kmol; and the pre-exponential factors  $D_{\rm o}$  being 3.37  $\times$   $10^{-6}$  and  $4.32 \times 10^{-6}$  m<sup>2</sup>/s for Imeco and Katowice, respectively) must be allowed to obtain a good quality fit of the experimental kinetic data obtained in the relevant range of reaction times and conditions for the practical application of the carbonate looping concept.

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# **Notation**

a, b = stoichiometric coefficients for carbonation reaction

 $C = \text{concentration of CO}_2, \text{ kmol/m}^3; \text{ b, bulk concentration; e,}$ equilibrium

D = effective product layer diffusivity, m<sup>2</sup>/s

 $D_{\rm o} = \text{pre-exponential factor in Eq. 10, m}^2/\text{s}$ 

 $D_{\rm p} = b M_{\rm CaO} D C / a \rho$  apparent product layer diffusion, m<sup>2</sup>/s

 $E_{\rm aK}$  = activation energy for the kinetic regime, K, kJ/kmol

 $E_{\rm aD}$  = activation energy for the combined diffusion and kinetic regime, D, kJ/kmol

 $h_{L, K-D}$  = product layer thickness, m; L, limiting product layer thickness; K-D transition between reaction regimes

 $k_{\rm so} = \text{pre-exponential factor in Eq. 7, m}^4/\text{kmols}$ 

 $k_{\rm s}$  = rate constant for surface reaction, m<sup>4</sup>/kmols

 $L_{o, N} = \text{initial total length of pore system, m/m}^3$ ; N, pore length for cycle N

 $M_{\text{CaO}} = \text{molecular weight of CaO, kg/kmol}$ 

 $R_i$  = pore radius, m

 $S_{o, N} = initial$  reaction surface, m<sup>2</sup>/m<sup>3</sup>; N, reaction surface for cycle N

t = time, s

 $V_{\mathrm{CaO,CaCo3}}^{\mathrm{M}}=\mathrm{molar}$  volumes, m³/kmol  $X=\mathrm{CaO}$  molar conversion; N, relative to cycle N; K-D, transition between reaction regimes; r, residual CaO conversion

Z = ratio volume fraction after and before reaction

### Greek letters

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\alpha= volume fraction of CaCO3 invading a pore volume \beta=2k_{\rm s}a\rho(1-\epsilon)/M_{\rm CaO}bD_{\rm p}S_{\rm o} \delta= decrease in pore radius upon carbonation, m \epsilon= porosity v_{\rm o}(r)= pore radii distribution \rho_{\rm CaO}= CaO density, kg/m³ \tau=k_{\rm s}(C_{\rm b}-C_{\rm c})S_{\rm o}t/(1-\epsilon) \Psi=4\pi L_{\rm o}(1-\epsilon)/S_{\rm o}^2
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